# A Compact Tensor Regression Model for Accelerating First-Principle Molecular Dynamics Simulation

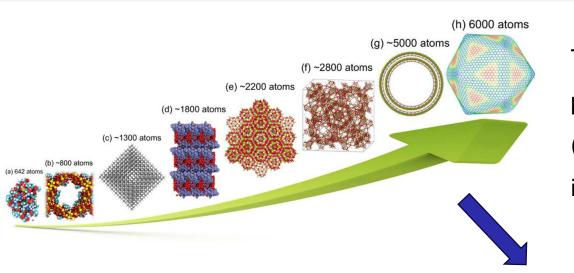
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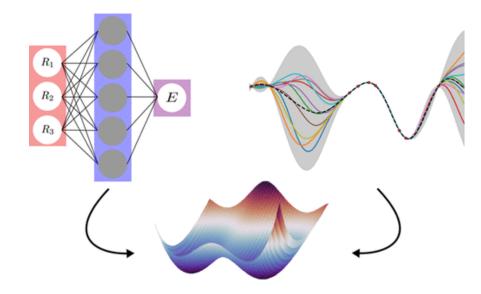
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## Background



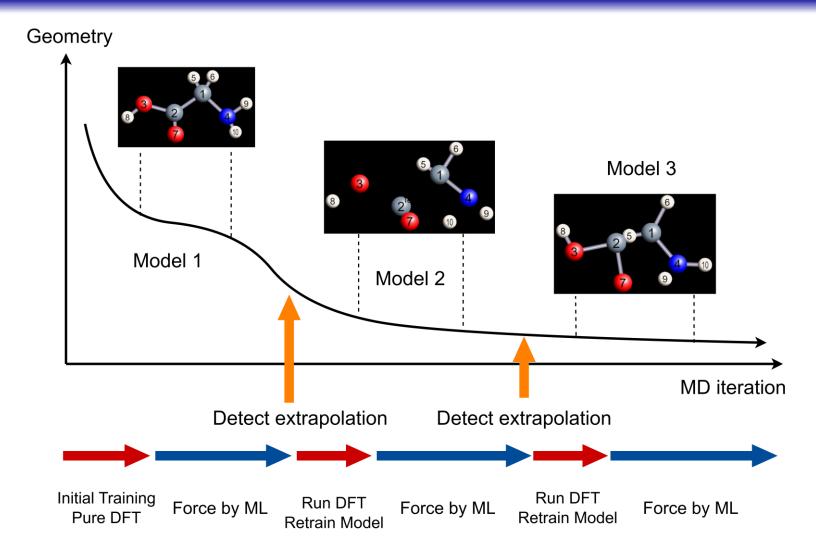
The desired system size for firstprinciples molecular dynamics (FPMD) simulation is largely increased.

Machine learning techniques has been proven to fit potential energy surface<sup>1</sup>. For better performance, researchers seek to develop **more complex model**, like deep neural network.



<sup>&</sup>lt;sup>1</sup> K. T. Butler et al., Nature 559, 547 (2018).

## Overview of on-the-fly fitting (active learning)



Model is not fixed by pre-train, also updated along MD simulation.

## Purpose of study

#### Main purpose

We try to fit the local PES for each atom with a compact atomic decomposed model and on-the-fly update along MD simulation.

#### **Highlight features**

- Fitting by solving linear equations problem (numerical robustness and low retrain cost)
- Independent model for each atom (O(N) time complexity and ability for complex chemical environment)
- Training data can be iteratively accumulated (Cold start and online training)

## Tensor representation of atomic energy

Total energy is represented by summation of atomic energy:

$$E_{\text{tot}}^{\text{model}} = \sum_{i} E_{i}^{\text{model}}$$

Them we consider the **atomic energy** for atom i:

$$n-1$$
 tiems

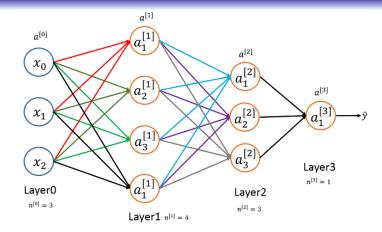
$$E_i^{\text{model}} = \langle V_i \left| \left( \left( A^{(n)} | V_i \rangle \right) | V_i \rangle \right) \dots | V_i \rangle \right),$$

$$A^{(n)}|V_i\rangle = \sum_j A^{(n)}_{i,j,k...,n} v_j = A^{(n-1)},$$

Where  $A^{(n)}$  is a **n-rank tensor**,  $|V_i\rangle$  is descriptor vector. For 1<sup>st</sup> and 2<sup>nd</sup> rank case:

$$E_i(A^{(1)}) = A_1 v_1 + A_2 v_2 + \dots + A_m v_m,$$
  

$$E_i(A^{(2)}) = E_i(A^{(1)}) + A_{22} v_1 v_1 + A_{23} v_1 v_2 + \dots$$



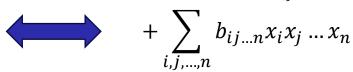
$$E = \sum_{i} \omega_{i} \sum_{j} \omega_{j} f_{j} \left( \sum_{k} f_{k}(\omega_{k} x_{k}) \right),$$

With **Taylor expansion** of f:

$$f_k(\omega_k x_k) = f(x_0) + f'(x_0)(\omega_k x_k - x_0) + \frac{1}{2!}$$
$$f''(x_0) (\omega_k x_k - x_0)^2 + \cdots$$

So, we have:

$$E = b_0 + \sum_i b_i x_i + \sum_{i,j} b_{ij} x_i x_j + \cdots$$



## Descriptor vector

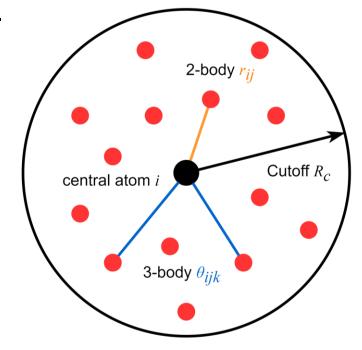
We consider the atomic energy is contributed by 2-

body and 3-body interactions in a finite range.

$$|V\rangle=(1,V_1^{rad},\ldots,V_p^{rad},V_1^{ang},\ldots,V_p^{ang})$$

$$V_p^{rad} = \sum_{j,j \in neighbor} h(Z_j) G_p^{rad}(r_{ij})$$

$$V_p^{ang} = \sum_{j,k;j,k \in neighbor} h(Z_j, Z_k) G_p^{ang}(\theta_{ijk})$$

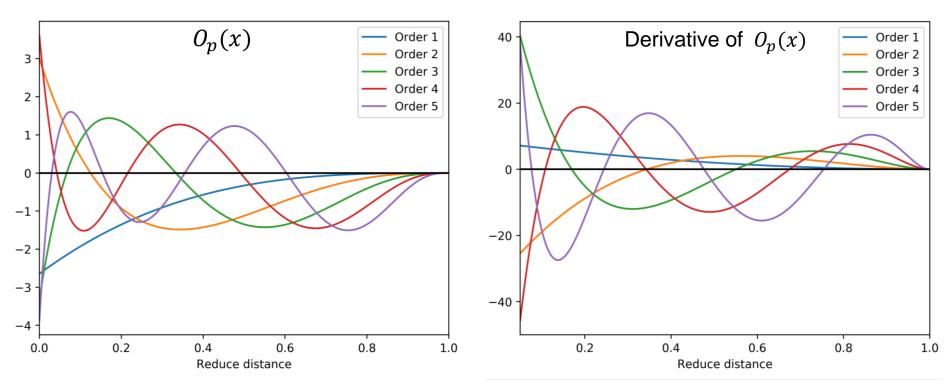


We use series of symmetry function  $G_p$  to **sample** all many body pairs inside cutoff range.  $Z_j$  is the normalized atomic number for distinguish **element** type.

## Orthogonal polynomial descriptor

#### Orthogonal polynomials with $(x-1)^p$ basis

The  $O_p(x) = \text{Orth}[(x-1)^p]$ ,  $p \ge 3$  series polynomial is forced to be orthogonal.



 $O_p$ ,  $O'_p$ ,  $O''_p$  should converge to 0 for **smooth energy change respect to neighboring atom change**.

### Loss function

The fitting process is to determine the tensor element in  $A_i^n$  by minimizing the loss function:

$$L = \sum_{N} \left( E_{N}^{\text{model}} - E_{N}^{\text{DFT}} \right)^{2} + \sum_{N} \sum_{axis} \left( F_{N,axis}^{\text{model}} - F_{N,axis}^{\text{DFT}} \right)^{2} + \lambda_{2} \sum_{N} ||A_{i}^{n}||$$

Thus, to minimize the loss function we can let derivative to 0. (2<sup>nd</sup> rank tensor case):

$$\frac{\partial L}{\partial A_{m'n'}} = \sum_{N} \left( E_{N}^{\text{model}} - E_{N}^{\text{DFT}} \right) \frac{\partial E_{N}^{\text{model}(i)}}{\partial a_{i,j,p}} + \sum_{N} \sum_{axis} \left( F_{N,axis}^{\text{model}} - F_{N,axis}^{\text{DFT}} \right) \frac{\partial F_{N,axis}^{\text{model}(i)}}{\partial A_{m'n'}} + 2\lambda_{2} \sum_{N} A_{m'n'}$$

Then the fitting process could be done alone the MD simulation and solving the

#### linear equations problem:

$$\left[\frac{\partial L}{\partial A_{m'n'}}\right] [A_{m'n'}]^T = 0$$

## **Implementation**

#### **Decouple atomic force**

$$F_{i,axis}^{\text{model}} = \frac{\partial E_{tot}^{\text{model}}}{\partial R_{i,axis}} = \frac{\partial E_{i}^{\text{model}}}{\partial R_{i,axis}} + \sum_{j \neq i} \frac{\partial E_{j}^{\text{model}}}{\partial R_{i,axis}} = F_{\text{self}}(A_{i,N}) + F_{\text{neighbor}}(A_{j,N})$$

 $F_{\text{neighbor}}(A_{j,N})$  contains parameters from neighboring atom which **can not solved independently**. So, we use the parameters  $A_{j,N-1}$  to approximate  $F_{\text{neighbor}}(A_{j,N})$  and use **autoregressive model to minimize residuals**.

$$F_{\text{diff}}(A_{j,N}, A_{j,N-1}) = \alpha_1 F_{\text{diff}}(A_{j,N-1}, A_{j,N-2}) + \alpha_2 F_{\text{diff}}(A_{j,N-2}, A_{j,N-3}) + \cdots$$

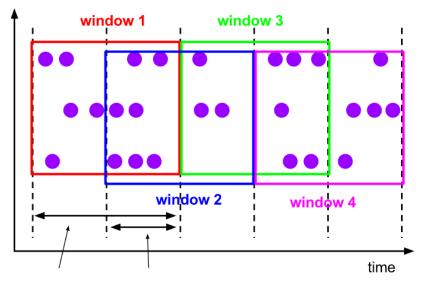
## **Implementation**

#### **Online training**

Cause the dimension of parameter tensor  $A_i^n$  is fixed, thus we can iteratively accumulate the training data by:

$$\frac{\partial L}{\partial A_{m'n'}} = A_{m'n'}^n = \omega * A_{m'n'}^{n-1} + A_{m'n'}^{\text{new}}$$

 $\omega$  is a **forgetting ratio** which is little smaller than 1 to **discard history step information** that is too far away from current configuration.



Force residuals given by autoregressive model reflect **long-time scale interaction** in MD trajectory. The idea is same in **recurrent neural networks** (RNN) and **graph neural networks** (GNN).

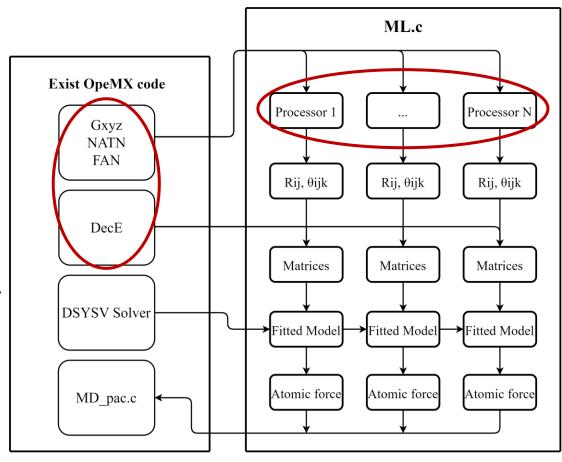
## Workflow with OpenMX

#### Scheme of ML code and OpenMX<sup>1</sup>

#### Fully parallelized for each atom

Necessary data from OpenMX

- Gxyz atom position
- FNAN neighboring atoms number
- NATN
   neighboring atoms serial number
- DecE decomposed energy

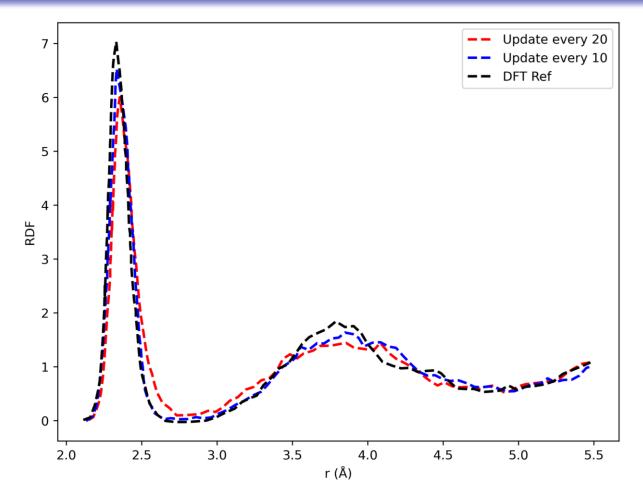


## Benchmark calculation of various systems

Isolated System	Energy MAE	Force MAE
Methene (CH <sub>4</sub> )	0.08 meV/atom	7 meV/Å
Nitrobenzene (C <sub>6</sub> H <sub>5</sub> NO <sub>2</sub> )	0.73 meV/atom	68 meV/Å
Sialic Acid (C <sub>11</sub> H <sub>19</sub> NO <sub>9</sub> )	1.07 meV/atom	114 meV/Å
Adenosine Triphosphate (C <sub>10</sub> H <sub>16</sub> N <sub>5</sub> O <sub>13</sub> P <sub>3</sub> )	1.32 meV/atom	145 meV/Å
Bulk System	Energy MAE	Force MAE
2D Graphene	0.23 meV/atom	27 meV/Å
Aluminum	0.47 meV/atom	76 meV/Å
Amorphous Silica (SiO <sub>2</sub> )	0.92 meV/atom	127 meV/Å

The typical accuracy range for ML potential is ~2 meV/atom and ~200 meV/Å.

## Radial Distribution Function

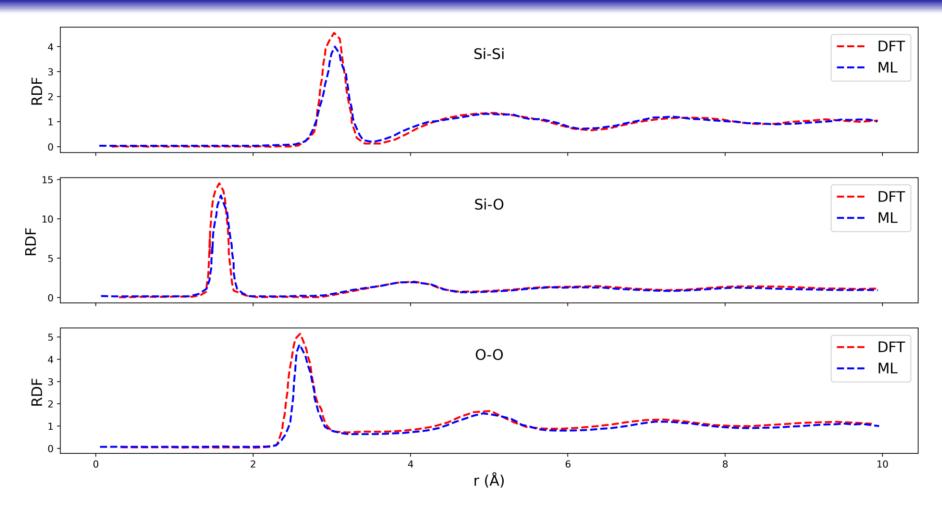


Two peaks are precisely reproduced by our tensor regression model.

RDF of amorphous silicon Si-Si distance under 300K.

(2000 fs in total, 400 fs for initial training, update every 10 steps).

## Radial Distribution Function



RDF of amorphous silica under 300K.

(2000 fs in total, 500 fs for initial training, update every 10 steps).

#### Conclusion

- We developed effective model of fitting local atomic energy surface, the accuracy is comparable to other ML potentials. Also, it able to reproduce the long-time scale statistical property.
- The accelerate ratio is around 10 by simple on-the-fly algorithm but lower than existing active learning method ( $\sim 10^3$ ).
- Our model shows enlightenment in the development of future machine learning potentials. Compact models which based on the physical system also possible for ML potentials compare with the more complex and data-driven large models.

## **Future Prospects**

- Design on-the-fly algorithm based on our model, for example the analysis of changes in atomic chemical environment.
- Application for electrode surface, because there are many possible surface structures and a long AIMD time is required for screening.
- Extending the model prediction capabilities to more complex highentropy alloy surfaces that always face reconfiguration during dynamics.
- Our final target system is electrochemical interface systems which are still challenge for existing ML potentials and DFT packages.

## Thank you for your listening.

## Density functional theory

Many-body Schrödinger equation

$$\widehat{H}\Psi(\boldsymbol{r},\boldsymbol{R}) = E\Psi(\boldsymbol{r},\boldsymbol{R})$$

With Born-Oppenheimer approximation, the electronic motion and the nuclear motion in molecules can be **separated**.

$$\Psi(\mathbf{r}, \mathbf{R}) = \phi_{ele}(\mathbf{r}) \chi_{nuc}(\mathbf{R})$$

Electron Schrödinger equation in Hartree unit

$$\left[-\frac{1}{2}\sum_{i}\nabla_{i}^{2}+\frac{1}{2}\sum_{i\neq j}\frac{1}{\left|\boldsymbol{r_{i}}-\boldsymbol{r_{j}}\right|}-\sum_{i,I}\frac{Z_{I}}{\left|\boldsymbol{r_{i}}-\boldsymbol{R_{I}}\right|}\right]\phi_{ele}(\boldsymbol{r})=E\phi_{ele}(\boldsymbol{r})$$

## Density functional theory

#### Hohenberg-Kohn theorem I

For any system of interacting particles in an external potential  $V_{ext}(\mathbf{r})$ , the potential  $V_{ext}(\mathbf{r})$  is determined uniquely by the ground-state particle density  $n_0(\mathbf{r})$ .

#### Hohenberg-Kohn theorem II

A *universal functional* for the energy E[n] can be defined in terms of the density. The exact ground state is the global minimum value of this functional.

With Hohenberg-Kohn theorem, the solution of the system ground state can be shifted from the solution of the wave function to the electron density. The system degrees of freedom are **reduced from 3N to 3**.

## Density functional theory

#### Kohn-Sham DFT

In Kohn-Sham scheme constructed an auxiliary system which the atoms are all independent of each other, and in turn we can solve the **single-electron Schrödinger equation**.

$$E[n] = -\frac{\nabla^2}{2} + \int d\boldsymbol{r} \, n(\boldsymbol{r}) V_{ext}(\boldsymbol{r}) + \frac{1}{2} \int d\boldsymbol{r} d\boldsymbol{r}' \, \frac{n(\boldsymbol{r}) n(\boldsymbol{r}')}{|\boldsymbol{r} - \boldsymbol{r}'|} + E_{xc}(n)$$

With Hohenberg-Kohn theorem II, the electron density that minimizes E[n] is the real electron density of the system. By the Lagrange multiplier method we can obtain the KS equation.

$$\left(-\frac{\nabla^2}{2} + V_{ext}(\mathbf{r}) + V_H(\mathbf{r}) + \frac{\delta E_{xc}[n]}{\delta n}\right) \psi_i(\mathbf{r}) = \varepsilon_i \psi_i(\mathbf{r})$$

After determining the form of  $E_{xc}[n]$ , KS equation can be solved by self-consistent field method.

## Linear scaling DFT

The core idea of the O(N) method is based on the assumption that the electron density n is given by the summation over each site i.

$$n^{\sigma}(\mathbf{r}) = \sum_{i\alpha,j\beta} \chi_{i\alpha}(\mathbf{r}) \chi_{j\beta}(\mathbf{r}) \rho^{\sigma}_{i\alpha,j\beta} = \sum_{i} (\sum_{\alpha,j\beta} \chi_{i\alpha}(\mathbf{r}) \chi_{j\beta}(\mathbf{r}) \rho^{\sigma}_{i\alpha,j\beta}) = \sum_{i} n^{\sigma}_{i}(\mathbf{r}),$$

The density matrix  $\rho$  is calculated by solving following linear equations:

$$H^{(i)}c_{\mu}^{(i)} = \epsilon_{\mu}^{(i)}S^{(i)}c_{\mu}^{(i)}.$$

where,  $H^{(i)}$  is the local Hamiltonian matrix,  $S^{(i)}$  is the overlap matrix. By truncating system the time complexity is reduced:

$$O(N^3) \to N \times O(N_{constant})$$

Further, the  $N_{constant}$  can be reduced by introducing Krylov subspace method, The Krylov subspace method used in OpenMX is a combination of DC and recursive methods to improve computational efficiency and robustness.

Because the pseudo-atomic orbitals  $\chi$  for pseudopotential in OpenMX is in finite range, the total energy can be decomposed into contributions from each atomic site i and localized orbital  $\alpha$ :

$$E_{\text{tot}} = \sum_{\sigma = \uparrow, \downarrow} \sum_{\nu} f_{\nu} \langle \phi_{\nu}^{\sigma} | \hat{T} | \phi_{\nu}^{\sigma} \rangle + \sum_{\sigma \sigma'} \int \omega_{\sigma \sigma'} n_{\sigma' \sigma} + \frac{1}{2} \iint \frac{n'(r)n'(r')}{|r - r'|} dv dv'$$

$$+ E_{\text{xc}} \{ n_{\sigma \sigma'} \} + E_{\text{cc}}$$

$$= E_{\text{band}} - \frac{1}{2} \int n' V_{\text{H}} dv - \int \text{Tr}(V_{\text{xc}} n) dv + E_{\text{xc}} + E_{\text{cc}}$$

$$= \sum_{i\alpha} \left[ \sum_{j\beta,\sigma\sigma'} \rho_{i\alpha,j\beta}^{\sigma\sigma'} F_{j\beta,i\alpha}^{\sigma'\sigma} \right] + \sum_{i\alpha j\beta} E_{\text{cc}}^{(i\alpha j\beta)}$$

$$= \sum_{i\alpha} E_{i\alpha},$$

The total energy in OpenMX is organized by:

$$E_{\text{tot}} = E_{\text{kin}} + E_{\text{na}} + E_{\text{NL}} + E_{\text{ec}} + E_{\delta \text{ee}} + E_{\text{xc}} + E_{\text{scc}}.$$

Kinetic energy

$$E_{kin} = \sum_{\sigma} \sum_{i\alpha} \left( \sum_{j\beta} \sum_{n}^{N} \rho_{\sigma,i\alpha,j\beta}^{(\mathbf{R_n})} h_{i\alpha,j\beta,kin}^{(\mathbf{R_n})} \right)$$
$$= \sum_{\sigma} \sum_{i\alpha} E_{\sigma,i\alpha,kin}$$

with

$$h_{i\alpha,j\beta,kin}^{(\mathbf{R_n})} = \frac{1}{V_B} \int_{BZ} dk^3 \sum_{\mu}^{Occ} \langle \psi_{\sigma\mu}^{(\mathbf{k})} | \hat{T} | \psi_{\sigma\mu}^{(\mathbf{k})} \rangle$$

#### Electron-core Coulomb energy (Non-local part)

$$E_{ec}^{(NL)} = \sum_{\sigma} \sum_{i\alpha} \left( \sum_{n}^{N} \sum_{j\beta} \rho_{\sigma i\alpha j\beta}^{(\mathbf{R_n})} h_{i\alpha,j\beta,NL}^{(\mathbf{R_n})} \right)$$

$$= \sum_{\sigma} \sum_{i\alpha} \left( \sum_{n}^{N} \sum_{j\beta} \rho_{\sigma i\alpha j\beta}^{(\mathbf{R_n})} \langle \phi_{i\alpha} | \sum_{I} V_{NL,I} | \phi_{j\beta} \rangle \right)$$

$$= \sum_{\sigma} \sum_{i\alpha} E_{\sigma i\alpha}^{(NL)}$$

#### Neutral atom energy

$$E_{na} = \int dr n(r) V_{na,I}$$

$$= \sum_{\sigma} \sum_{i\alpha} \left( \sum_{j\beta} \sum_{n}^{N} \rho_{\sigma,i\alpha,j\beta}^{(\mathbf{R_n})} h_{i\alpha,j\beta,na}^{(\mathbf{Rn})} \right)$$

$$= \sum_{\sigma} \sum_{i\alpha} E_{\sigma,i\alpha}^{na}$$

#### Screened core correction energy

$$E_{scc} = \frac{1}{2} \sum_{I,J} \left( \frac{Z_I Z_J}{|\tau_I - \tau_J|} - \int dr n_I^{(a)}(r) V_{H,J}^{(a)}(r) \right)$$

$$= \sum_{\sigma} \sum_{i\alpha} \left( \frac{1}{2N_i} \sum_j \frac{Z_I Z_J}{|\tau_I - \tau_J|} \right)$$

$$- \int dr n_I^{(a)}(r) V_{H,J}^{(a)}(r)$$

$$= \sum_{\sigma} \sum_{i\alpha} E_{\sigma,i\alpha}^{scc}$$

#### Electron-electron Coulomb energy

$$E_{\delta ee} = \frac{1}{2} \int dr \left( n(r) - \sum_{I} n_{I}^{(a)}(r) \right) \delta V_{H}(r)$$

$$= \sum_{\sigma} \sum_{i\alpha} \frac{1}{2} \left( \sum_{j\beta} \sum_{n}^{N} \rho_{\sigma,i\alpha,j\beta}^{(\mathbf{R_{n}})} h_{i\alpha,j\beta}^{\delta V} - \frac{1}{2} \int dr \frac{n_{i}^{(a)}(r)}{N_{i}} \delta V_{H}(r) \right)$$

$$= \sum_{\sigma} \sum_{i\alpha} E_{\sigma,i\alpha}^{\delta ee}$$

#### Exchange correlation energy

$$E_{xc} = \int dr (n(r) + n_{pcc}(r)) \epsilon_{xc}(r)$$

$$= \sum_{\sigma} \sum_{i\alpha} \left( \sum_{n}^{N} \sum_{j\beta} \rho_{\sigma,i\alpha,j\beta,xc}^{(\mathbf{R_n})} h_{i\alpha,j\beta,xc}^{(\mathbf{R_n})} + \frac{1}{2N_i} \int dr n_{pcc,i}(r) \epsilon_{xc}(r) \right)$$

$$= \sum_{\sigma} \sum_{i\alpha} E_{\sigma,i\alpha}^{xc}$$